Influence of point defects on the metal-insulator transition in BaVS\(_3\)

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To investigate the nature of the metal-insulator transition which takes place in BaVS\(_3\) at \(T_{\text{MIT}}=70\) K, we study pure samples of BaVS\(_3\) irradiated by fast electrons, which introduce point defects. Our results show that the metal-insulator transition is gradually weakened and shifted to lower temperatures by introducing the defects, in close analogy to systems where metal-insulator transition is due to a Peierls mechanism. Additionally, we study the influence of point defects on the temperature-pressure phase diagram of BaVS\(_3\).

Here, we report an ambient and high-pressure transport study of BaVS\(_3\) where point defects were induced by fast electron irradiation. We find that by introducing interstitials and vacancies by knock-on collisions, the MI transition is progressively smeared out and shifted to lower temperatures. Under high pressure, the \(T_{\text{MIT}}\) line in the phase diagram is pushed toward lower pressures.

I. INTRODUCTION

BaVS\(_3\) is a quasi-one-dimensional itinerant spin-1/2 system, built from VS\(_3\) chains spaced by barium atoms. The vanadium 3\(^d\) electron is equally shared between two bands, a wide quasi-one-dimensional \(d_z\) and a narrow isotropic \(e_g\) band.\(^1\)\(^2\) Despite the chainlike crystal structure and the quasi-one-dimensionality, BaVS\(_3\) is electrically nearly isotropic.\(^3\) It is believed that this is due to the fact that electron correlations strongly reduce the on-chain conductivity. At ambient pressure, the compound undergoes three transitions: a structural transition from hexagonal to orthorombic symmetry at \(T_S=240\) K, a metal-insulator transition at \(T_{\text{MIT}}\approx70\) K, and a magnetic transition at \(T_N=30\) K, where an incommensurate antiferromagnetic order is established. By diffuse x-ray scattering, it was observed that a commensurately tetramerized structure develops below \(T_{\text{MIT}}\), suggestive of a Peierls transition in the quasi-one-dimensional \(d_z\) band.\(^4\) However, no charge disproportionation larger than 0.01\(e^-\) was found below \(T_{\text{MIT}}\) by anomalous scattering at vanadium \(K\) edge.\(^5\)

Therefore, it is still not firmly established what is the mechanism of the metal-insulator (MI) transition. In many classical charge or spin density wave systems, it has been studied how the radiation damage influences the phase transition. A similar response of the metal-insulator transition in BaVS\(_3\) to point defects may suggest that its origin is in the long-range density wave formation. Irradiation by a beam of fast electrons can be used to produce defects in a controlled way. Studies of several systems featuring a Peierls phase transition have shown that point defects locally perturb the coherency of the static charge ordering and may ultimately prevent the system from long-range ordering.

The presence of defects is also related to an interesting issue in BaVS\(_3\): its extreme sensitivity to disorder, seen in several of its physical properties.\(^6\) For example, sulfur content has a very important influence on the properties of BaVS\(_3\). Strongly sulfur-deficient samples are known to exhibit no MI transition and to have a ferromagnetic ground state.\(^7\) Moreover, the structural transition occurring at \(T_S\) can be detected in the resistivity only in the samples with a stoichiometric sulfur content. Still, no systematic study of how the defects influence the MI transition, and the phase diagram has been investigated so far.

II. METHODS

The crystals of BaVS\(_3\) were grown in tellurium flux and annealed in sulfur atmosphere at high temperature.\(^8\) Several long needlelike crystals were cut perpendicular to the chain direction. In this study, we use six samples, coming from two single crystals. The starting crystals are of high quality, the structural transition occurring at \(20\) K, where an incommensurate antiferromagnetic order is established. By diffuse x-ray scattering, it was observed that a commensurate tetramerized structure develops below \(T_{\text{MIT}}\), suggestive of a Peierls transition in the quasi-one-dimensional \(d_z\) band.\(^4\) However, no charge disproportionation larger than 0.01\(e^-\) was found below \(T_{\text{MIT}}\) by anomalous scattering at vanadium \(K\) edge.\(^5\)

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III. RESULTS AND DISCUSSION

The top panel of Fig. 1 shows the temperature dependence of the resistivity for a single sample repetitively treated by electron radiation. The measurements of resistivity are per-
formed in situ, at ambient pressure, directly after irradiating the sample and annealing it for a brief interval of several minutes at 100 K. The defects produced by irradiation are fairly sensitive to thermal migration, which is why the temperature range of the measurement is limited to 20–80 K. The resistivity of the sample above the MI transition increases under irradiation. The lower panel of Fig. 1 shows the logarithmic derivative \( d \left( \frac{\log \rho}{\log T} \right) / d(1/T) \) in the same temperature range. The transition at 70 K in the pure sample shifts to lower temperatures and is significantly widened by the presence of defects. In fact, before the \( T_{\text{MI}} \) is decreased by 10 K, the metal-insulator transition is already completely smeared out. The broadening and the shift of the transition under the influence of defects have been seen in several systems that undergo a charge-density wave (CDW) transition.\(^9\) Such behavior is reminiscent of a Peierls mechanism driving the metal-insulator transition. Because the defects pin the phase of the density wave, coherence between the chains is weakened. The transition is thus shifted to lower temperatures and gradually suppressed. The decrease in \( T_{\text{MI}} \) is monotonous and approximately linearly dependent on electron fluence, as can be seen in Fig. 2.

The resistivity at temperatures below the MI transition decreases as the electron dose is increased. After irradiating the sample by a fluence of \( 3.14 \times 10^{19} \) e\(^{-}\)/cm\(^2\), the resistivity at 35 K decreases by more than 20 times. There are two reasons for the lowered resistivity. First, the defects introduce localized states through which electrons can hop, enhancing the conductivity of the insulating phase. Second, the semiconducting gap is weakened because the pinning effect decreases the coherence volume of the CDWs. The value of the semiconducting gap decreases as the irradiation dose grows from the initial value of \( \sim 530 \) to \( 440 \) K after the dose of \( 3.14 \times 10^{19} \) e\(^{-}\)/cm\(^2\). Here, the values of gap are extracted from the temperature dependence of the resistivity between 50 and 30 K using a fit to the activated behavior,

\[
\rho = \rho_0 \exp \left( \frac{\Delta}{k_B T} \right).
\]

A similar disorder-induced decrease in the gap has previously been observed in several CDW systems.\(^9\)

When the samples of BaVS\(_3\) are irradiated by the flux of fast electrons at the temperature of 20 K, various scattering processes may take place, resulting in displacements in all the three atomic sublattices. The cross sections for Ba, V, and S atoms depend on the value of the threshold energy \( E_d \), which is the minimum energy necessary to eject an atom from its site. Typical values of \( E_d \) range from 10 to 30 eV depending on the structure and material. For example, \( E_d \) values for O and Cu displacements in YBa\(_2\)Cu\(_3\)O\(_{7-\delta}\) are, respectively, 10 and 15 eV.\(^{10}\) As BaVS\(_3\) is a rather open structure, we may suppose that the values of \( E_d \) are relatively low. The number of defects created per VS\(_3\) unit is \( n_d \), where \( \sigma_i^{\phi} \) is the cross section for displacing the atom \( i \) from its site and \( \phi \) is the electron fluence. Calculations of \( n_d \) by taking, for instance, \( E_d=5 \) eV for S and 15 eV for V, lead to a total number of displaced V and S atoms per VS\(_3\) plane of \( n_d=2.0 \times 10^{-2} \) for an electron fluence of \( 10^{19} \) e\(^{-}\)/cm\(^2\). Here, we do not consider the defects in the Ba sublattice because \( E_d \) for Ba is presumably large, and moreover, these defects would not be expected to have such an important influence.

FIG. 1. (Color online) The temperature dependence of resistivity (top panel) and its logarithmic derivative for the same single crystal irradiated by different electron fluences. The sample was kept below 100 K, except in the case of the last two curves where the annealing time and temperature are indicated in the parentheses.

FIG. 2. (Color online) The dependence of the transition temperature on the electron fluence during the irradiation of four different samples. The samples were maintained on temperatures below 100 K.
onto the physics of BaVS\textsubscript{3} as S and V defects. Finally, even though this absolute value of \( n_d \) may not be very accurate, it allows a comparison between the different crystals studied.

Annealing the sample at room temperature drastically reduces the concentration of impurities. From a sample irradiated by \( 3.14 \times 10^{19} \, e^-/cm^2 \), shown in Fig. 1, where the transition is practically wiped out, the annealing at 300 K for several days recovers the phase transition at a temperature with only a slightly less sharp peak in the logarithmic derivative and merely \( \sim 1 \) K lower than in the pure sample. This indicates that a vast part of the defects are easily recombined through thermal diffusion. The peak in the resistivity derivative of the \( 3.14 \times 10^{19} \, e^-/cm^2 \) curve is positioned between the peaks of the curves measured \textit{in situ} after irradiation fluences of \((0.19–0.3) \times 10^{19} \, e^-/cm^2 \). Using the above estimates of displacement cross sections, this would correspond to \( n_d \sim 5 \times 10^{-3} \), i.e., 0.5% V and S displacements per unit formula. Since the sulfur atoms are expected to thermally migrate easier than the vanadium atoms, when the sample is annealed by warming up to room temperature, it is likely that the majority of the remaining defects are in the V sublattice.

Another way of estimating the number of defects which remain after electron irradiation and a subsequent annealing at 300 K is through a measurement of magnetic susceptibility. We have investigated three pieces of the starting single crystal, irradiated, respectively, by the total electron fluences of \( 3.3 \times 10^{19} \), \( 1.6 \times 10^{19} \), and \( 0.0 \, e^-/cm^2 \). The irradiated samples showed a decrease in the \( T_{MI} \) of 1.9 and 0.8 K, respectively. For a pure BaVS\textsubscript{3} sample, the temperature dependence of magnetic susceptibility shows a \( 1/T \) behavior down to \( T_{MI} \), a sharp antiferromagneticlike drop below the MI transition, and a negligible Curie tail below \( \sim 15 \) K.\(^3\) The presence of defects does not significantly influence the high-temperature part, but it considerably increases the low-temperature Curie contribution. The magnitude of the Curie tail due to the presence of defects can be evaluated using the following formula:

\[
\chi = \frac{N_{def} \mu_{eff}^2}{T}.
\]  

Here, \( N_{def} \) is the density of defects and \( \mu_{eff} \) is their effective magnetic moment. If we assume that the majority of the defects in the samples are due to spin-1/2 cations \( V^{4+} \) and spin-1/2 anions \( S^- \), we obtain that the effective spin-only moment is \( \mu_{eff} = 1.73 \mu_B \). Using the above formula, we may estimate that the density of defects corresponding to a fluence of \( 10^{19} \, e^-/cm^2 \) and a subsequent annealing at room temperature for several days approximately equal 0.5% of defects per unit formula. The latter result is approximately a factor of 3 of the estimate we got from the above calculation using the cross sections for V and S atom displacements, which implies that there may be other non-negligible contributions to the Curie tail.

As we have seen, the MI transition in BaVS\textsubscript{3} can be completely suppressed by introducing the point defects. Another way to do so is to apply hydrostatic pressure to the sample. The temperature dependence of the resistivity under various pressures, for a pristine sample and a sample with irradiation-produced point defects, is shown in the upper panel of Fig. 3. The external pressure gradually diminishes nesting, destroying the tetramerization along the chains, which leads to a decrease in \( T_{MI} \) and a complete suppression of the insulating phase under \( p_{cr} \approx 2.0 \) GPa.\(^{11}\)

The lower panel of Fig. 3 shows the phase diagram of BaVS\textsubscript{3} in the presence of defects, in comparison to a pure sample. We observe that the \( T_{MI} \) phase boundary is shifted to lower pressures and lower temperatures as the number of defects increases. The \( p_{cr} \) for a sample irradiated by \( 6.27 \times 10^{19} \, e^-/cm^2 \) is 0.3 GPa lower than for the pristine sample. We believe that the reason there is a decrease in \( T_{MI} \) is the same as for the ambient pressure: adding the defects pins the phase of the CDW in the \( d_{z^2} \) band, which weakens the coherence between the chains. The semiconducting gap is related to the transition temperature through the mean field relation \( \Delta = mk_B T_{MI} \). They are both determined by the energy gain which results from lowering the electronic state energy due to the lattice deformation. If the three-dimensional coherence volume of the CDWs is reduced by pinning effects on defects, the electronic energy gain is diminished and so is the Peierls gap. In case of BaVS\textsubscript{3}, \( m = 12 \) and seems to be approximately independent of pressure (up to \( \sim 1.8 \) GPa) (Ref. 12) and of defect concentration.

The most interesting part of the effect of point defects on the phase diagram is that close to \( p_{cr} \). In the pristine sample, the insulating phase rapidly collapses above 1.7 GPa.\(^{13}\) This collapse is attributed to the strong interaction of the \( d_{z^2} \) and
shape of the resistivity indicates that there may be a change in the energy dispersion of a collective mode responsible for the conduction electron scattering in this temperature range. Since at such high temperatures the electronic scale is expected to give less temperature dependence, a remaining possibility is that the introduced defects alter the phononic energy spectrum. For example, if point defects broaden the phonon dispersion, by modifying the elastic constants, then there may be more phonons available for electron-phonon scattering already at lower temperatures than in the pristine sample. This, in turn, lowers the temperature where a term linear in temperature appears in the resistivity. By Raman spectroscopy, two possible candidates were found, $\sim 193$ and $\sim 350$ cm$^{-1}$ modes, which appear in the temperature range relatively close to room temperature. Finally, such a phononic interpretation of the high-temperature part of the resistivity is in accord with our rather elevated estimate of the defect density of the order of 1%.

IV. CONCLUSION

We have shown that the pinning of the density wave to the defects gradually destroys the three-dimensional order. With $\sim 5\%$ of defects, the metal-insulator transition is entirely suppressed. Furthermore, the influence of defects on the phase diagram is such that the $T_{\text{MI}}$ is lowered and its sudden drop above 1.7 GPa becomes less pronounced. Finally, the strong influence of the defects on the thermal dependence of the resistivity at high temperature suggests that the point defects may infer changes to the phononic spectrum.

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